Introduction to Total Scattering and Reverse Monte Carlo Methods

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ISIS Neutron and Muon Source

Neutrons for Chemistry and Materials Science Applications

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Outline

- About me.
- Introduction to total scattering.
- Instrument requirements for total scattering/PDF methods.
- Modelling total scattering data.
- Scientific case studies.
 - A closer look at RMCProfile.
 - **Conclusions & discussion points**





About Me

I am a materials chem

- MChem at Warwick Un

 Final year project on h
- PhD in Chemistry at Wa
 - Hydro/solvothermal s
 - Structural refinement
 - Total scattering metho
- Postdoctoral fellowship
 - Structure and propert
- Postdoctoral researche
 RMCProfile developm
- Instrument scientist at

 Responsible for total s



rument scientist.

oxides

& Dr. Alex Hannon) terials

d Polaris



Local and Average Viewpoints

Average viewpoint:



Local viewpoint:









Information contained in a diffraction pattern:

- Size and shape of unit cell (peak positions)
- Symmetry within the unit cell (absences)
- Contents of the unit cell (relative intensities)
- Thermal motion
- Particle size
- Strain
- Texture



What is total scattering?



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What is total scattering?



What is total scattering?



What is total scattering?



What is total scattering?



The pair distribution function (PDF)

A histogram of pairwise interatomic distances produced by Fourier transformation of the total scattering function.

Differential
cross section
$$\frac{1}{N} \frac{d\sigma}{d\Omega} = \sum_{i,j=1}^{n} c_i c_j \bar{b}_i \bar{b}_j [A_{ij}(Q) - 1] + \sum_{i=1}^{n} c_i \bar{b}_i^2$$

$$F(Q) = \sum_{i,j=1}^{n} c_i c_j \bar{b}_i \bar{b}_j [A_{ij}(Q) - 1]$$
Pair distribution function
or PDF
$$G(r) = \frac{1}{(2\pi)^3 \rho_0} \int_0^\infty 4\pi Q^2 F(Q) \frac{\sin Qr}{Qr} dQ$$



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)



The pair distribution function (PDF)





This is the neutron PDF for quartz-type SiO_2 (measured on Polaris).







Measure diffraction data to a high maximum Q

- the spatial resolution of the PDF depends on Q_{max}
- need high energy (short wavelength) X-rays or spallation neutrons































Choose the Q-range that is right for YOUR sample



Choose the Q-range that is right for YOUR sample



Measure diffraction data to a high maximum Q

- the spatial resolution of the PDF depends on Q_{max}
- need high energy (short wavelength) X-rays or spallation neutrons

Use an instrument with good reciprocal space resolution

broadened Bragg peaks result in a dampened PDF





The effect of Q-space resolution

Higher resolution in reciprocal space = less damping in PDF



Measure diffraction data to a high maximum Q

- the spatial resolution of the PDF depends on Q_{max}
- need high energy (short wavelength) X-rays or spallation neutrons
- Use an instrument with good reciprocal space resolution
 - broadened Bragg peaks result in a dampened PDF

Maximise signal-to-noise ratio

- brighter sources
- lots of detectors
- long (enough) measurement times


Maximising signal-to-noise ratio





- New, brighter neutron sources offer a much greater flux of neutrons.
- They will only be useful for total scattering if the right instruments are built!
- SNS has NOMAD, also second target station coming?
- J-PARC has NOVA.
- ESS ???



Maximising signal-to-noise ratio

- Less bright sources can maximise count-rate by using as many detectors as possible around the sample.
- This also increases the d/Q range accessible, BUT...
- ...the instrument resolution
 varies with 2θ and flight path.
- Which makes data processing somewhat more complicated!



GEM Since 1999



Polaris upgrade Since 2012



Maximising signal-to-noise ratio



- It is not surprising that shorter measurements produce noisier data.
- How much of a problem this is will depend on the specifics of the sample and experiment!



Measure diffraction data to a high maximum Q

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Maximise signal-to-noise ratio

- brighter sources
- lots of detectors
- long (enough) measurement times

Measure all background contributions carefully

non-sample scattering must be removed



In an ideal situation we would have a sample floating in "mid-air", but in most cases this isn't going to be possible...





- Removing "background" is easy if the sample is in a thin vanadium can or TiZr cell.
- It is much more difficult when the environment is more complicated, e.g. in situ cells or the Paris Edinburgh press.
- More, and longer, background measurements will be required.
- SE design should seek to minimise background contributions.
- Data quality may always be compromised?









Minimising and removing background scattering



- These data are from several grams of CeO₂ in a glass tube.
- Although CeO₂ Bragg peaks are clearly visible, the total scattering is dominated by the tube!



Measure diffraction data to a high maximum Q

- the spatial resolution of the PDF depends on Q_{max}
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Use an instrument with good reciprocal space resolution

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Maximise signal-to-noise ratio

- brighter sources
- lots of detectors
- long (enough) measurement times

Measure all background contributions carefully

• non-sample scattering must be removed

Process the data carefully and appropriately

- put the data on an absolute scale
- understand instrument geometry and calibration





Going from raw counts to F(Q)

- Generally for diffraction analysis, the raw spectra are grouped into "banks" of similar scattering angle.
- What about total scattering?





The dangers of incorrect calibration



The dangers of incorrect calibration

Modelling total scattering data



Introduction



Modelling total scattering data

Modelling techniques

There are two main ways in which detailed structural information can be extracted from total scattering data: <u>small box</u> and big box modelling.



Small box modelling:

- Crystal structure refined to fit the PDF: "real-space Rietveld"
- Limited to crystallographic descriptions of structural parameters.
- Identify discrepancies between average and local structure.



Modelling total scattering data

Modelling techniques

There are two main ways in which detailed structural information can be extracted from total scattering data: small box and <u>big box</u> modelling.



Big box modelling:

- Reverse Monte Carlo (RMC).
- Supercell of >10,000 atoms, moved at random to obtain best possible agreement with all data.
- Atomistic model that is consistent with input data.
- Not constrained by symmetry.







Atomic configuration

- This is a "box of atoms"
- Needs to be fairly large: 10,000 100,000 atoms
- Crystalline material:
 - supercell of the (refined) unit cell
 - careful if partial/mixed occupancy
 - check for unphysical distances
- Amorphous material:
 - distribute atoms randomly
 - molecular dynamics simulation
 - other possibilities?









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The power of randomness

- Without the constraints of symmetry, the RMC algorithm can explore more of phase space.
- This can be necessary to model certain types of systems.
- Take this doped CeO₂ as an example...



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Why accept some "bad" moves?

- If only "good" moves are accepted, you risk getting stuck in a local minimum.
- The probability of accepting a "bad" move depends on how much it worsens the goodnessof-fit.
- Moves that violate hard constraints are always rejected.



Time for a break!



new/novel porous or porosity explosive ga 203 conductor **neutron** amorphous battery zeolite metastable framework ceo2 gold in situ green PDF local inorganic batio3 silver multiferroic batio3 silver ceramic bizo3 magnetism inorganic multiferroic batio3 silver ceramic bizo3 magnetism bizo3 magnetism alloy sp nano* alloy sponge quantum Structure piezoelectric quantum Structure piezoelectric fuel cell niche spinel bacteria configuration glass high pressure organic complex liquid water crystalline X-ray simulation copper perovskite oxide distortion unusual molecular dynamics Reverse Monte Carlo semiconductor or functional average phase transition insight







A disordered polymorph of Ga₂O₃



- Potential photocatalyst and catalyst support
- Structure poorly understood
- Cubic spinel structure
- Rietveld refinement reveals four partially occupied Ga sites
- Nanocrystalline

H. Y. Playford, A. C. Hannon, E. R. Barney, and R. I. Walton, *Chem. Eur. J.*, 2013, **19**, 2803–2813.





- Small-box modelling of the PDF
- Medium-to-high *r* agrees well with average crystal structure
- Large discrepancies in local structure
- Improved fit when lower symmetry model is used, but it is a purely local effect



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Random starting model: Ga-Ga < 1Å Handmade 2x2x2 cell with reasonable distances: supercell with artificial superstructure **Re-randomised supercell** using atom swapping & fit to Bragg pattern Physically and chemically sound starting model(s) for full RMC refinement



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RMC refinement using 6x6x6 supercell

- vastly improved fit to local structure



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RMC refinement using 6x6x6 supercell

- vastly improved fit to local structure
- maintains correct average



Collapsed RMC box

Unit cell



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RMC provides bond length and angle distributions:

 these distributions are the sum of 200 refined "boxes of atoms"

the O_h sites are highly distorted

- the crystal structure defines two very different T_d sites
- but locally these sites are very similar









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Non-weighted Ga-O partials 10 $Ga_{8a}-O$ $Ga_{4af}-O$ $Ga_{4af}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ $Ga_{16c}-O$ RMC provides bond length and angle distributions:

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The data clearly show the octahedra are distorted, but what do they actually look like?

- multiple RMC runs provide ensemble of >700,000 polyhedra to analyse!
- 50% all 6 bonds shorter than the mean bond length
- 40% [3+3] type



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- multiple RMC runs provide ensemble of >700,000 polyhedra to analyse!
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- 40% [3+3] type

Thermodynamically stable β -Ga₂O₃ has [3+3] type...

```
Locally, cubic \gamma-Ga<sub>2</sub>O<sub>3</sub> = monoclinic \beta-Ga<sub>2</sub>O<sub>3</sub>
```



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A closer look at RMCProfile



- Implementation of the RMC algorithm particularly suited to crystalline materials.
- "Profile" refers to the Bragg profile a very important constraint for average structure.
- Based on the original RMCA code of McGreevy and Puzstai, extended by Matt Tucker (now at ORNL).
- Developers from many institutions including ISIS, ORNL, QMUL, Oxford, Cambridge, NIST, Chalmers...
- The program is available online at <u>www.rmcprofile.org</u>
- It can fit multiple datasets (X-ray and neutron PDF, F(Q), Bragg)...
- ...and use "chemical sense" in the application of appropriate constraints.





- Current version: RMCProfile 6.7.x
 - was developed at ISIS by Wojciech Slawinski
 - incorporation of various user requested features
 - improved usability

• RMCProfile 6.8

- final release of version 6 (coming soon)
- incorporation of developments from Igor Levin's team at NIST

• RMCProfile 7.0

- being developed at ISIS by Wojciech Slawinski
- big news: multiphase RMC (multiple 'boxes' of atoms)
- currently in need of input from users
- available now for interested beta testers!

• RMCProfile 7.1 and later

- speed and efficiency improvements
- developments for nanostructured systems
- improved support for X-ray data
- regular incremental updates



www.rmcprofile.org



A "live" demonstration

- Barium titanate, BaTiO₃
- Perovskite
- Surprisingly complex phase diagram!
- Neutron scattering data collected on Polaris

Unit cell

- Rietveld refinement using GSAS
- Total scattering data processed using GudrunN





A "live" demonstration





F(Q) initial fit

Bragg initial fit



A "live" demonstration





D(r) initial fit





```
😑 batio3_4hr.dat 🗵
   TITLE :: BaTiO3 4hr
 1
   MATERIAL :: BaTiO3
 2
 3 PHASE :: Tetragonal
 4 TEMPERATURE :: 293
                                 Number of partials =
 5 INVESTIGATOR :: HYP
                                       n(n+1)/2
 6
 7 MINIMUM DISTANCES :: 1.68 1.68 1.68 1.68 1.68 1.68 Angstrom
 8 MAXIMUM MOVES :: 0.05 0.05 0.05 Angstrom
 9 R SPACING :: 0.0200 Angstrom
10 PRINT PERIOD :: 1000
11 TIME LIMIT .: . . . . . 900.00 MINUTES
12
   SAVE PERIOD :: 10.00 MINUTES
13
14 INPUT CONFIGURATION FORMAT :: rmc6f
15
   SAVE CONFIGURATION FORMAT :: cfg
16
17
   IGNORE HISTORY FILE ::
18
19
   ATOMS :: Ba Ti O
20
   FLAGS ::
21
22
   > NO MOVEOUT
23 > NO SAVE CONFIGURATIONS
  > NO RESOLUTION CONVOLUTION
24
                                                                       ogy Facilities Council
2 5
```



l	25	
l	26	NEUTRON_REAL_SPACE_DATA :: 1
l	27	<pre>> FILENAME :: batio3_rmc_4hr_32.99.gr</pre>
l	28	> DATA_TYPE :: G(r)
l	29	> FIT_TYPE :: D(r)
l	30	> START_POINT :: 1
l	31	> END_POINT :: 1000
l	32	<pre>> CONSTANT_OFFSET :: 0.00</pre>
l	33	<pre>> WEIGHT ::</pre>
l	34	> NO_FITTED_OFFSET
l	35	> NO_FITTED_SCALE
l	36	
l	37	NEUTRON_RECIPROCAL_SPACE_DATA :: 1
l	38	<pre>> FILENAME :: batio3_rmc_4hr_32.99.fg</pre>
l	39	$ \cdot \cdot > \cdot DATA_TYPE \cdot : : \cdot F(Q) $
l	40	···>·FIT_TYPE·::··F(Q)
l	41	> START_POINT :: 1
l	42	> END_POINT :: 1620
l	43	> CONSTANT_OFFSET :: 0.00
l	44	<pre>> CONVOLVE ::</pre>
l	45	> WEIGHT :: 0.100
l	46	> NO_FITTED_OFFSET
l	47	> NO_FITTED_SCALE
l	48	
l	49	BRAGG ::
l	50	> BRAGG_SHAPE :: gsas3
l	51	> SUPERCELL :: 10 10 10
l	52	> RECALCULATE
	53	> DMIN :: 0.5
I	54	> WEIGHT :: 0.005
	55	
	56	END ::
۱	57	

68 68 69 DISTANCE_WINDOW :: 70 >> MNDIST :: 0 1.750 0 71 >> MXDIST :: 0 2.270 0







RMCProfile			×
C:\RMCProfile_package>cmd.exe/K			
C:\RMCProfile_package≻echo OFF C:\RMCProfile_package≻cd C:\Users\jjm23479\Desktop\Demo\batio3_demo			
C:\Users\jjm23479\Desktop\Demo\batio3_demo>rmcprofile batio3_4hr > log.txt			
		chnology Facili	ties Counci
	\bigcirc ISIS		

A "live" demonstration



- In most cases, the χ² drops quickly at first, and then gradually approaches convergence.
- In this case, convergence is reached within an hour or so...
- Let's have a look at the results (live!)



A "live" demonstration





F(Q) final fit

Bragg final fit



A "live" demonstration





A "live" demonstration





Requirements for a successful RMCProfile refinement:

- Quality data
- Multiple datasets
- Single phase sample
- Good powder average
- Average structure well-characterised
- Understanding of structural chemistry
- Targeted analysis of refined configurations



Why was the gallium oxide case study successful?



Why was the gallium oxide case study successful?

- Quality data
 - Data were from GEM, a TOF diffractometer optimised for TS measurements
 - The instrument was well calibrated
 - The data were of good statistical quality



Why was the gallium oxide case study successful?

- Quality data 🗸
- Multiple datasets

- Bragg diffraction pattern and TS data fitted simultaneously



Why was the gallium oxide case study successful?

- Quality data 🗸
- Multiple datasets
- Single phase sample 🗸



Why was the gallium oxide case study successful?

- Quality data 🗸
- Multiple datasets
- Single phase sample 🗸
- Good powder average



Why was the gallium oxide case study successful?

- Quality data
- Multiple datasets
- Single phase sample \checkmark
- Good powder average
- Average structure well-characterised

- I did a very thorough Rietveld investigation before beginning the PDF analysis



Why was the gallium oxide case study successful?

- Quality data
- Multiple datasets
- Single phase sample 🗸
- Good powder average ✓
- Average structure well-characterised \checkmark
- Understanding of structural chemistry \checkmark

- I knew what kinds of bonding environments Ga "preferred" and how close it could get to other atoms.



Why was the gallium oxide case study successful?

- Quality data
- Multiple datasets
- Single phase sample \checkmark
- Good powder average
- Average structure well-characterised \checkmark
- Understanding of structural chemistry \checkmark
- Targeted analysis of refined configurations \checkmark

- I wanted to understand how the octahedral sites were distorted, and my analysis reflected that.



And another thing...



Conclusions

- Total scattering is an extension of powder diffraction that changes the viewpoint from average to local.
- It is incredibly useful (and we have not yet found its limits) but it is NOT magic.
- The pair distribution function is simple and intuitive, but real structural information requires modelling!
- Reverse Monte Carlo refinements using RMCProfile produce atomistic models that are consistent with all available data
 - Requirements for success include prior characterisation and an understanding of what you want to know about your structure.

